

Oxygen Partial Pressure Dependent Electrical Conductivity of $\text{LiNb}_{1-x}\text{Ta}_x\text{O}_3$ Solid Solutions

Yuriy Suhak¹, Ahsanul Kabir¹, Dmitry Roshchupkin²,
Boris Red'kin², Steffen Ganschow³, and Holger Fritze¹

¹ Institute of Energy Research and Physical Technologies, Clausthal University of Technology, Am Stollen 19 B, 38640 Goslar, Germany

² Institute of Microelectronics Technology and High Purity Materials, Russian Academy of Science, Chernogolovka, Moscow District, 142432, Russia

³ Leibniz-Institut für Kristallzüchtung, Max-Born-Str. 2, 12489, Berlin, Germany
ahsanul.kabir@tu-clausthal.de

Summary:

The electrical conductivity is investigated for piezoelectric single crystals of congruent lithium niobite (LiNbO_3 , LN), lithium tantalate (LiTaO_3 , LT), and their solid solutions ($\text{LiNb}_{1-x}\text{Ta}_x\text{O}_3$, LNT) at varying oxygen partial pressure in the range of $0.2 \cdot 10^{-15}$ bar at a constant temperature of 900°C . These crystals have been grown by the Czochralski and micro-pulling down method. At sufficiently low oxygen partial pressure, the LN sample demonstrates about two orders of magnitude larger electrical conductivity than LT. Such an outcome indicates a dominant n-type conductivity in the former. Moreover, the increase of the Nb/Ta ratio shows a tendency to decrease conductivity for LNT.

Keywords: piezoelectric single crystal, lithium niobate-tantalate solid solutions, electrical conductivity, high temperature

Background, Motivation, and Objective

High temperatures sensors and actuators enable a wide range of advanced technological applications in aerospace, automobile, energy conversion, and beyond [1,2]. Considering materials potentially suited for the above-mentioned devices, piezoelectric single crystals with large electromechanical coupling factors, high melting/phase transition temperatures, and cost-effective growth methods are essential [3,4]. However, state-of-the-art material such as quartz ($\alpha\text{-SiO}_2$) is thermally unstable, showing a phase transition ($\alpha \rightarrow \beta$) at around 570°C . In contrast, the family of langasite crystal displays no phase transformation up to the melting point ($\sim 1400^\circ\text{C}$), but they possess a relatively low piezoelectric coefficient [2]. The ferroelectric lithium niobate (LN) and lithium tantalate (LT) have attracted significant interest because of their excellent dielectric, electro-optic, electro-acoustic, photoelectric, and piezoelectric properties [5]. Nevertheless, the high-temperature performance of these materials is restricted by thermo-chemical instability of LN and low curie temperature of LT. Meanwhile, few publications [6,7] and our preliminary work indicate that the properties of $\text{LiNb}_{1-x}\text{Ta}_x\text{O}_3$ (LNT) solid solutions can be tailored which includes refractive index, electroacoustic and electrooptic properties, melting point and curie temperature. In the current work, we investigate the oxygen partial pressure ($p\text{O}_2$) dependent high-temperature

electrical properties of LN and LT crystals prepared by two different methods. Additionally, such experiments are also performed with LNT with different Nb/Ta ratios.

Description of the Method

The crystals investigated in this experiment were produced by the Czochralski (CZ) and micro-pulling down ($\mu\text{-PD}$) technique at the Institute of Microelectronics Technology and High Purity Materials (Russia) and at the Leibniz Institute for Crystal Growth (Berlin), respectively. The CZ crystals were then cut into a disc-shaped dimension (diameter 10 mm and thickness 1 mm) with a Z-cut orientation and polished. The $\mu\text{-PD}$ crystals were grown along Z-axis with about 15 mm in length and 1 mm in diameter. The samples were partially coated with Pt paste via screen printing and thermally treated at 1000°C for 30 min. The electrical properties were measured by electrochemical AC impedance spectroscopy (Solartron 1260, UK) in a wide frequency range ($10^6\text{-}10^0$ Hz) under an alternative of bias of 100 mV at 900°C and changing oxygen partial pressure ($10^0\text{-}10^{-15}$ bar). The data were fitted using an equivalent circuit model consisting of a bulk resistor connected in parallel with a constant phase element. The conductivity was estimated using the equation, $\sigma = L/RA$ where R is the bulk resistance, A the electrode area, and L the sample thickness.

Results

The bulk electrical conductivity of LN and LT samples is represented in Fig. 1 as a function of p_{O_2} at 900 °C. As can be seen, the sample manufactured by the μ -PD method shows a higher electrical conductivity than the CZ sample. Such an outcome is expected due to a relatively high impurity content caused by rapid cooling during the μ -PD process. The latter prevents segregation and, thereby, lowering of impurity concentrations as expected for CZ growth. Further, LN possesses noticeably larger conductivity than LT, especially at low p_{O_2} . For LT, the conductivity remains p_{O_2} independent down to 10^{-9} bar and increases slightly for increasingly reducing atmosphere. In contrast, a strong p_{O_2} dependent conductivity is observed for LN below about 10^{-3} bar, suggesting the LN system has a lower reduction enthalpy than LT. In general, the enhancement of the conductivity is typical for a mixed ionic-electronic conductor which exhibits an increasing electronic (n-type) conductivity as the p_{O_2} reduces [4,8].

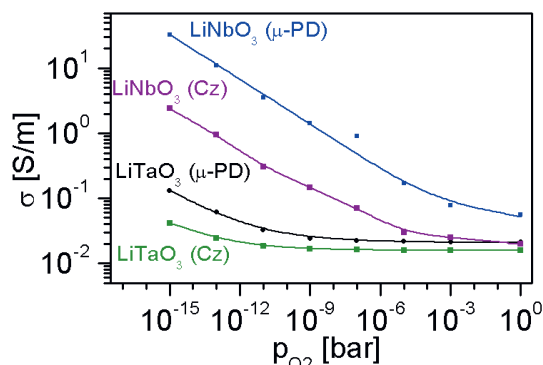


Fig. 1. Bulk electrical conductivity of LN and LT crystals grown by Czocharalski and micro-pulling down technique for varying oxygen partial pressure, measured at 900 °C.

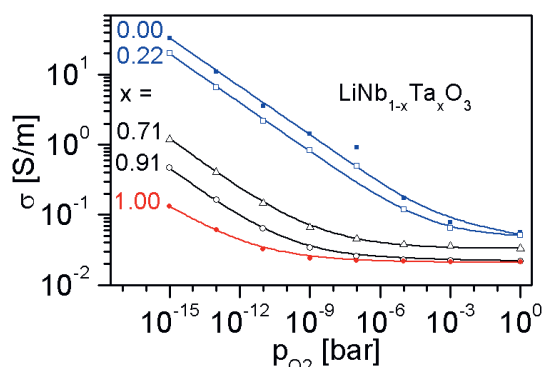


Fig. 2. Bulk electrical conductivity of LNT compounds with different Nb/Ta ratios prepared via micro-pulling down technique as a function of oxygen partial pressure, measured at 900 °C.

Fig. 2 illustrates the p_{O_2} -dependent bulk electrical conductivity of LNT for different Nb/Ta ratios. The Nb-rich composition shows an identical reduction tendency to pure LN. The electrical conductivity is progressively dropped with increasing Nb/Ta ratio, and expectedly, the Ta-rich sample shows largely suppression of reduction like LT.

Conclusions

In summary, the electrical conductivity of LN, LT, and LNT solid solution was examined at a constant temperature of 900 °C with regards to different oxygen partial pressures. The outcome reveals a significant rise in conductivity for the LN sample than the LT at a low-oxygen partial pressure. The electrical conductivity can be tuned by the Nb/Ta ratio.

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