

# Enhanced H<sub>2</sub>S Sensing Properties of Porous SnO<sub>2</sub> Nanofibers Modified with CuO

Yan Zhao, Xiuli He, Jianping Li, Xiaoguang Gao and Jian Jia

State Key Laboratory of Transducer Technology, Institute of Electronics, Chinese Academy of Sciences, Beijing, China, 100190

Corresponding email: [hxl@mail.ie.ac.cn](mailto:hxl@mail.ie.ac.cn)

## Abstract:

Porous SnO<sub>2</sub> nanofibers were fabricated by electrospinning and oxygen plasma treatment. Subsequently, CuO was sputtered on the porous SnO<sub>2</sub> fibers as the modifier. H<sub>2</sub>S sensing properties of the fibers were tested and the effect of CuO amount on the response and recovery characteristics to H<sub>2</sub>S was investigated. It was found that the response is strongly dependent on the CuO amount and the response time and recovery time become longer with the increase of the CuO amount. The highest response of the fibers to 2 ppm H<sub>2</sub>S attains  $1.9 \times 10^3$  at the operating temperature of 100 °C with the optimum CuO thickness of 24 nm, and the fibers show good selectivity to H<sub>2</sub>S.

**Key words:** SnO<sub>2</sub> nanofibers, electrospinning, CuO-modified, H<sub>2</sub>S sensing

## Introduction

Sensing materials based on p-type CuO and n-type SnO<sub>2</sub> have been reported showing high sensitivity and good selectivity to H<sub>2</sub>S due to the formation of p-n junctions in air and the disruption of the p-n junctions by the conversion of CuO to CuS upon exposure to H<sub>2</sub>S [1, 2]. Over the past decades, enormous efforts have been made to fabricate different kinds of H<sub>2</sub>S sensing materials based on nanosized SnO<sub>2</sub> films and CuO films or particles by different methods [3-9]. However, the compacted films obstruct gas diffusion and adsorption and are not the best choice for gas sensing materials.

In recent years, one-dimensional nanomaterials, such as nanofibers, nanowires, nanoribbons and nanorods have been widely investigated for gas detection, and the sensitivity can be improved and response time can be shortened due to the high surface to volume ratio [10] and quantum confinement effect [11]. Hence considerable efforts have been made to fabricate one-dimensional H<sub>2</sub>S sensing materials via thermal oxidation, thermal evaporation, hydrothermal process and electrospinning [10, 12, 13]. Xue et al. synthesized core/shell p-n junctions with hydrothermal SnO<sub>2</sub> nanorods as core and CuO nanoparticles as shell [10]. This core/shell junctions exhibited a high sensitivity about  $9.4 \times 10^2$  and short response time about 30 s to 10 ppm H<sub>2</sub>S at operating temperature of 60 °C. Kong et al. [12] fabricated CuO modified SnO<sub>2</sub>

nanoribbons with thermal oxidation followed by mixing with CuO powders. It is reported that the sensor has high sensitivity (~180) and short response time (~15 s) to 3 ppm H<sub>2</sub>S at room temperature. Vivek et al. obtained CuO doped SnO<sub>2</sub> nanowires by thermal evaporation [13]. The nanowires doped with proper concentration of CuO were reported exhibiting high sensitivity about  $6 \times 10^6$  to 50 ppm H<sub>2</sub>S at 150 °C.

In the present work, a novel method to fabricate CuO-modified porous SnO<sub>2</sub> nanofibers was proposed. Porous SnO<sub>2</sub> nanofibers were fabricated by electrospinning, followed by plasma treatment and annealing, and CuO were sputtered on SnO<sub>2</sub> nanofibers. The effects of the CuO amount on H<sub>2</sub>S sensing properties of the prepared nanofibers were investigated.

## Experimental

The electrospinning solution was prepared by dissolving 0.5 g polyvinylpyrrolidone (PVP, Sigma-Aldrich Trading Corp. Ltd., Mw=1.3×10<sup>6</sup> g/mol) and 1 g SnCl<sub>4</sub>·5H<sub>2</sub>O in 4 mL DMF (N, N-dimethylformamide, Beijing Chemical Reagent Corp.) and 6 mL ethanol, and then sufficiently stirring at room temperature. The flow rate of the solution was 0.1 mL/h and the electrode-to-collector distance was settled at 4 cm. The optimized applied voltage was 8 kV. The fibers were deposited on a silicon based substrate which was embedded with a serpentine Pt heater and Pt interdigital electrodes [14].

The as-prepared fibers were dried for 4 h at 80 °C to ensure that the solvents were completely vaporized. Afterward, the fibers were treated with oxygen plasma of 0.28 W/cm<sup>2</sup> for 20 minutes at room temperature to modify the surface of fibers and form porous structure [14, 15]. After annealing at 660 °C for 4 h in ambient environment, SnO<sub>2</sub> nanofibers with porous structure were obtained.

CuO was deposited on porous SnO<sub>2</sub> nanofibers by rf magnetic sputtering. The Cu metal target (50 mm in diameter and 99.99% purity) was adopted and the substrate was kept at room temperature. The sputtering was carried out at a working gas pressure of 0.8 Pa in a mixture of Ar and O<sub>2</sub> (7: 3) with rf power of 50 W. The deposition rate is 18 nm/min and the thickness can be controlled by the deposition time. Finally, the modified fibers were annealed in air at 500 °C for 4 h with the ramp of 2 °C/min. The morphology of the fibers were examined by scanning electron microscopy (SEM, HITACHI S-4800), and the crystal structure of SnO<sub>2</sub> and CuO were examined by X-ray diffraction (XRD) which was conducted on a Philips X'pert Pro X-ray diffractometer with Cu K $\alpha$ 1 radiation ( $\lambda$ =1.5406 Å).

Gas sensing properties were measured using a static test system and current is employed to characterize the resistance of fibers. Dry synthetic air was used as both a reference gas and a diluting gas to obtain desired concentrations of H<sub>2</sub>S. A certain volume H<sub>2</sub>S with the concentration of 525 ppm was injected into the test chamber by a syringe through a rubber plug. After fully mixed with the diluting gas, the fibers were put into the test chamber. When the response reached a constant value, the fibers were taken out to recover in dry air. H<sub>2</sub>S concentrations less than 0.5 ppm were obtained by secondary dilution. The response is defined as  $I_g/I_a$ , where  $I_a$  and  $I_g$  are the currents of the fibers upon exposure to dry air and H<sub>2</sub>S, respectively. The response time is specified as the time to reach 90% of the maximum change of current after H<sub>2</sub>S is injected. The recovery time is defined as the time to return to 10% of the maximum change of current after the removal of H<sub>2</sub>S.

## Results and discussion

The XRD pattern in Fig. 1 reveals all the major diffraction peaks of the SnO<sub>2</sub> and CuO after annealing. All the diffraction peaks can be perfectly indexed as the tetragonal rutile structure SnO<sub>2</sub> (JCPDS 41-1445) and the monoclinic structure CuO (JCPDS 05-0661), respectively.

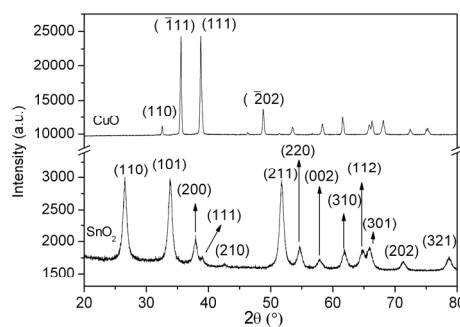


Fig. 1. XRD patterns of SnO<sub>2</sub> nanofibers and CuO after annealing

In the work, samples of CuO-modified porous SnO<sub>2</sub> nanofibers with the CuO deposited for 0 s, 20 s, 40 s, 60 s, 80 s and 100 s, corresponding to the thickness of 0, 6 nm, 12 nm, 18 nm, 24 nm and 30 nm, are marked as 1<sup>#</sup>, 2<sup>#</sup>, 3<sup>#</sup>, 4<sup>#</sup>, 5<sup>#</sup> and 6<sup>#</sup>, respectively.

Fig. 2(a) shows the SEM image of the porous SnO<sub>2</sub> nanofibers. It can be seen that the fibers show coarse morphologies with small particles and holes on the surface. During the oxygen plasma treatment, PVP was removed, resulting in a porous microstructure. The formation of the porous structure was proposed in our previous works [14, 15]. The SEM images of CuO-modified SnO<sub>2</sub> nanofibers with different CuO deposition times are shown in Fig. 2(b-f). It is clear that the porous structure of the fibers is maintained but the porosity decreases with the increase of the CuO deposition time due to the superfluous CuO.

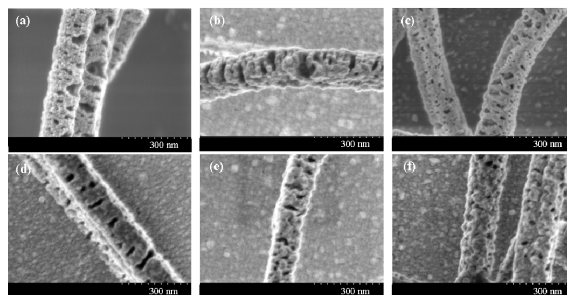


Fig. 2. The SEM images of CuO-modified porous SnO<sub>2</sub> nanofibers with the CuO deposition time of (a) 0 s, (b) 20 s, (c) 40 s, (d) 60 s, (e) 80 s and (f) 100 s.

Fig. 3 shows the response of CuO-modified porous SnO<sub>2</sub> nanofibers with different CuO deposition time towards 2 ppm H<sub>2</sub>S as the function of operating temperature. It can be seen that pure SnO<sub>2</sub> nanofibers are almost insensitive to H<sub>2</sub>S at the operating temperature ranging from 30 °C to 170 °C. However, the responses of CuO-modified SnO<sub>2</sub> nanofibers increase with a raise of operating temperature, and attain the maximum value at about 100 °C, then decrease with further raise of the operating temperature. In addition, it is apparent in Fig. 3 that the response of CuO-modified SnO<sub>2</sub>

nanofibers is strongly dependent on the CuO amount, and it reaches the highest response (about  $1.9 \times 10^3$ ) to 2 ppm  $\text{H}_2\text{S}$  with the optimum CuO thickness of 24 nm, then decreases quickly as the CuO thickness increased. The enhanced performance in CuO-modified porous  $\text{SnO}_2$  nanofibers can be understood in terms of the formation of p-CuO/n- $\text{SnO}_2$  junctions on the surface of the fibers in air and disruption of junctions by the sulfurization of CuO into conductive CuS upon exposure to  $\text{H}_2\text{S}$ , which cause a large decrease in electrical resistance of the fibers [16].

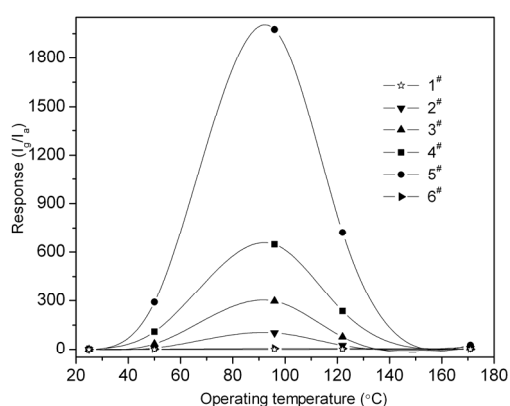


Fig. 3. The response of the samples as a function of operating temperature.

The sensing characteristics of  $\text{SnO}_2$  nanofibers modified with different amount of CuO upon exposure to 2 ppm  $\text{H}_2\text{S}$  at optimal operating temperature of  $100^\circ\text{C}$  were examined and the results are illustrated in Tab. 1. The slowdown of the response/recovery is observed with the increase of CuO deposition time.

Tab. 1: Response time and recovery time of the samples to 2 ppm  $\text{H}_2\text{S}$  at  $100^\circ\text{C}$ .

Sample number	1 <sup>#</sup>	2 <sup>#</sup>	3 <sup>#</sup>	4 <sup>#</sup>	5 <sup>#</sup>	6 <sup>#</sup>
Response time / s	-	50	65	100	115	180
Recovery time / s	-	22	22	26	30	40

Fig. 4(a) shows the response/recovery of sample 5<sup>#</sup> to  $\text{H}_2\text{S}$  from 0.01 ppm to 4 ppm at  $100^\circ\text{C}$ . The inset shows the response/recovery of the fibers to low concentration  $\text{H}_2\text{S}$  in magnified view. It is obvious that the current increase upon exposure to  $\text{H}_2\text{S}$ , and decrease rapidly and return to the original value after  $\text{H}_2\text{S}$  is removed. The response is about  $0.9 \times 10^3$ , and the response time and recovery time are about 105 s and 25 s upon exposure to 1 ppm  $\text{H}_2\text{S}$ , respectively. With the increase of  $\text{H}_2\text{S}$

concentration, the response time became longer. The response of the fibers as a function of  $\text{H}_2\text{S}$  concentration when operated at  $100^\circ\text{C}$  is shown in Fig. 4(b). For  $\text{H}_2\text{S}$  concentration of 0.01 ppm, the response is about 1.2 and the detection limit is less than 10 ppb. Moreover, the response rapidly increases with increasing of  $\text{H}_2\text{S}$  concentration, and there is an approximate linear relationship between the response and the concentration in the range of 0.01 ppm - 4 ppm.

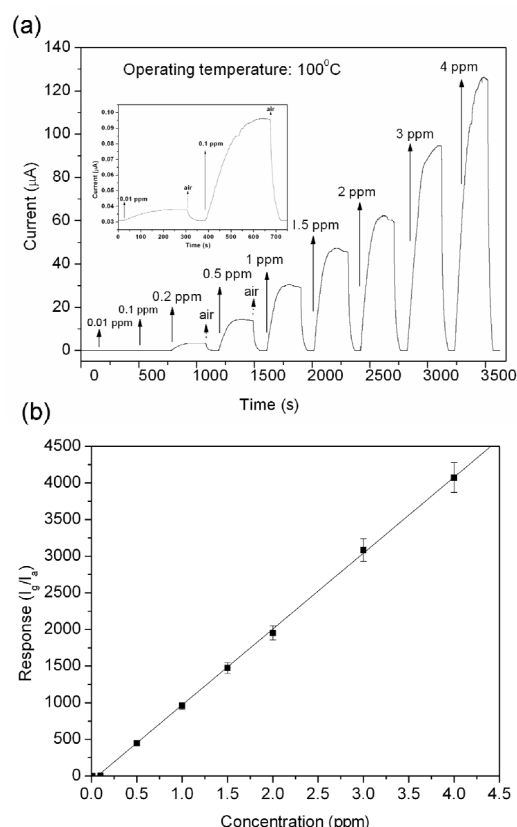


Fig. 4. (a) The response/recovery of sample 5<sup>#</sup> to  $\text{H}_2\text{S}$  in the range of 0.01 ppm to 4 ppm at  $100^\circ\text{C}$ , (b) the response as a function of  $\text{H}_2\text{S}$  concentration.

To investigate the selectivity of CuO-modified porous  $\text{SnO}_2$  fibers to  $\text{H}_2\text{S}$ , the response of the fibers to interference gases including  $\text{CH}_3\text{CH}_2\text{OH}$ ,  $\text{SO}_2$ ,  $\text{NO}$  and  $\text{CH}_4$  have been investigated. Fig. 5 depicts the response of sample 5<sup>#</sup> to 2 ppm  $\text{H}_2\text{S}$  and other pollutant gases with the concentration of 10 ppm at  $100^\circ\text{C}$ . The fibers are totally insensitive to interference gases and it shows that the porous  $\text{SnO}_2$  nanofibers modified with optimal amount of CuO exhibit prominent selectivity to  $\text{H}_2\text{S}$ . The good selectivity of the fibers is attributed to the disruption of CuO- $\text{SnO}_2$  junctions upon exposure to  $\text{H}_2\text{S}$ .

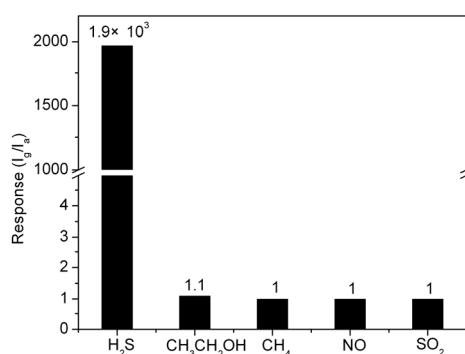


Fig. 5. The selectivity of sample 5<sup>#</sup> to H<sub>2</sub>S.

## Conclusion

A novel method to fabricate H<sub>2</sub>S sensing materials based on CuO-modified porous SnO<sub>2</sub> nanofibers was explored. The porous SnO<sub>2</sub> nanofibers were fabricated with electrospinning followed by oxygen plasma treatment and annealing, then CuO was sputtered on the surface of the porous SnO<sub>2</sub> nanofibers. The CuO-modified fibers exhibit high response to low concentration H<sub>2</sub>S at the operating temperature of 100 °C, and the responses varied with the amount of CuO. The highest response of 1.9 × 10<sup>3</sup> is obtained to 2 ppm H<sub>2</sub>S at CuO thickness of 24 nm, and the response time and recovery time is about 115 s and 30 s. Furthermore, CuO-modified porous SnO<sub>2</sub> fibers show good selectivity to H<sub>2</sub>S. The enhanced performance of the porous fibers is primarily due to the porous structure of fibers which was helpful to form CuO-SnO<sub>2</sub> junctions as many as possible in the surface of fibers.

## Acknowledgements

This work is supported by the National Natural Science Foundation of China (grant no. 60871055).

## References

- [1] T. Maekawa, J. Tamaki, N. Miura, N. Yamazoe, Sensing behavior of CuO-loaded SnO<sub>2</sub> element for H<sub>2</sub>S detection, *Chemistry Letters* 4, 575-579 (1991); doi: 10.1246/cl.1991.575.
- [2] S. Manorama, G.S. Devi, V.J. Rao, Hydrogen sulfide sensor based on tin oxide deposited by spray pyrolysis and microwave plasma chemical vapor deposition, *Applied Physics Letters* 64, 3163-3165 (1994); doi: 10.1063/1.111326.
- [3] A. Khanna, R. Kumar, S. S. Bhatti, CuO-doped SnO<sub>2</sub> thin films as hydrogen sulfide gas sensor, *Applied Physics Letters* 82, 4388-4390 (2003); doi: 10.1063/1.1584071.
- [4] J.P. Li, Y. Wang, X.G. Gao, Q. Ma, L. Wang, J.H. Han, H<sub>2</sub>S sensing properties of the SnO<sub>2</sub>-based thin films, *Sensors and Actuators B* 65, 111-113 (2000); doi: 10.1016/S0925-4005(99)00406-2.
- [5] V.R. Katti, A.K. Debnath, K.P. Muthe, M. Kaur, A.K. Dua, S.C. Gadkari, S.K. Gupta, V.C. Sahni, Mechanism of drifts in H<sub>2</sub>S sensing properties of SnO<sub>2</sub>: CuO composite thin film sensors prepared by thermal evaporation, *Sensors and Actuators B* 96, 245-252 (2003); doi: 10.1016/S0925-4005(03)00532-X.
- [6] A. Chowdhuri, P. Sharma, V. Gupta, K. Sreenivas, H<sub>2</sub>S gas sensing mechanism of SnO<sub>2</sub> films with ultrathin CuO dotted islands, *Journal of Applied Physics* 92, 2172-2180 (2002); doi: 10.1063/1.1490154.
- [7] A. Chowdhuri, V. Gupta, K. Sreenivas, Fast response H<sub>2</sub>S gas sensing characteristics with ultra-thin CuO islands on sputtered SnO<sub>2</sub>, *Sensors and Actuators B* 93, 572-579 (2003); doi: 10.1016/S0925-4005(03)00226-0.
- [8] A. Chowdhuri, V. Gupta, K. Sreenivas, R. Kumar, S. Mozumdar, P. K. Patanjali, Response speed of SnO<sub>2</sub>-based H<sub>2</sub>S gas sensors with CuO nanoparticles, *Applied Physics Letters* 84, 1180-1183 (2004); doi: 10.1063/1.1646760.
- [9] J.H. Jeun, S.H. Hong, CuO-loaded nano-porous SnO<sub>2</sub> films fabricated by anodic oxidation and RIE process and their gas sensing properties, *Sensors and Actuators B* 151, 1-7 (2010); doi: 10.1016/j.snb.2010.10.002.
- [10] X. Y. Xue, L. L. Xing, Y. J. Chen, S. L. Shi, Y. G. Wang, and T. H. Wang, Synthesis and H<sub>2</sub>S sensing properties of CuO-SnO<sub>2</sub> core/Shell PN-junction nanorods, *Journal of Physical Chemistry C* 112, 12157-12160 (2008); doi: 10.1021/jp8037818.
- [11] Z.L. Wang, Zinc oxide nanostructures: growth, properties and applications, *Journal of Physics: Condensed Matter* 16, 829-858 (2004); doi: 10.1088/0953-8984/16/25/R01.
- [12] X. H. Kong, Y. D. Li, High sensitivity of CuO modified SnO<sub>2</sub> nanoribbons to H<sub>2</sub>S at room temperature, *Sensors and Actuators B* 105, 449-453 (2005); doi: 10.1016/j.snb.2004.07.001.
- [13] V. Kumara, S. Sen, K. P. Muthe, N. K. Gaur, S. K. Gupta, J. V. Yakhmi, Copper doped SnO<sub>2</sub> nanowires as highly sensitive H<sub>2</sub>S gas sensor, *Sensors and Actuators B* 138, 587-590 (2009); doi: 10.1016/j.snb.2009.02.053.
- [14] X. L. He, Y. Zhao, J. P. Li, X. G. Gao, J. Jia, Porous CuO electrospun fibers and their gas sensing properties, *Sensor Letters* 9, 294-298 (2011); doi: 10.1166/sl.2011.1467.
- [15] Y. Zhang, J. P. Li, G. M. An, X. L. He, Highly porous SnO<sub>2</sub> fibers by electrospinning and oxygen plasma etching and its ethanol-sensing properties, *Sensors and Actuators B* 144, 43-48 (2010); doi: 10.1016/j.snb.2009.10.012.
- [16] A.W. Tang, S.C. Qu, K. Li, Y.B. Hou, F. Teng, J. Cao, Y.S. Wang Z.G. Wang, One-pot synthesis and self-assembly of colloidal copper(I) sulfide nanocrystals, *Nanotechnology* 21, 285602 (2010); doi: 10.1088/0957-4484/21/28/285602.